



Catalytic combustion of methane on $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ oxides



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ABSTRACT

A series of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ ($x=0\text{--}0.5$) perovskite oxides were prepared by a sol-gel method. X-ray diffraction spectrometer (XRD), BET surface area measurements, scanning electron microscopy (SEM) images, and temperature-programmed reduction (TPR) were used to characterize their physical structures and redox properties. Catalytic methane combustion tests for $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ ($x=0\text{--}0.5$) perovskite oxides show that the activity of LaFeO_3 was highly improved due to the introduction of Ce in the A-site of the perovskite catalysts. Among all the catalysts, $\text{La}_{0.7}\text{Ce}_{0.3}\text{FeO}_3$ has the maximum oxidative performance with the corresponding T_{90} as low as 510 °C. Combining with density functional theory calculation, it was suggested that the electrons of Fe ions increase in $\text{La}_{0.875}\text{Ce}_{0.125}\text{FeO}_3$ due to the introduction of Ce^{4+} ion, which leads to stronger interactions with adsorbed O_2 . Correspondingly, the adsorption energy of O_2 on $\text{La}_{0.875}\text{Ce}_{0.125}\text{FeO}_3$ increases and the O–O bond is activated. Thus, the Ce doped perovskite has higher oxidative activity than pure LaFeO_3 .

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1. Introduction

Catalytic combustion of methane has received much attention due to its increasing application in environment-friendly fuel [1]. To eliminate methane in tail gas of natural gas automobile, a catalyst with high performance is necessary. Noble metals supported on oxide carriers are the good choices [2,3]. Whereas with the depleting resources and increasing prices of noble metals, a series of non-noble metal-based catalysts have been explored. The perovskite-type oxides (ABO_3) are the most promising alternatives to partially substitute the noble metal-based catalysts due to their unique structures and high performances in catalytic oxidation [4–6]. A perovskite-type oxide has an ABO_3 crystal structure wherein A-site cation with a large ionic radius has 12 coordination to oxygen atoms, and B-site cation with a smaller ionic radius has six coordination. The most outstanding advantage of perovskite oxides is their wide variety of composition and constituent elements with the basic structure unchanged, which make them widely apply into electronic and catalytic materials [7,8].

Since the first report in catalytic hydrocarbon combustion by Libby [9], a series of LaBO_3 has been extensively studied as catalysts to eliminate automobile exhaust. Blasin-Aubé et al. [10] studied the catalytic oxidation of a variety of VOCs with different functional groups over $\text{La}_{0.8}\text{Sr}_{0.2}\text{MnO}_{3+x}$, and found that the total

oxidation temperature was below 623 K. The acetone, isopropanol, and benzene oxidation performances of LaMnO_3 and LaCoO_3 catalysts were systematically compared by Spinicci et al. [11]. Based on the investigation of LaMO_3 ($M=\text{Mn, Fe, Cu, Co, Ni, Cr}$) oxides in methane combustion, Arai et al. [12] found that the partially cation-substituted La–Fe, La–Mn, and La–Co oxide systems showed high oxidation activities. An excellent review by Tejuca et al. [6] pointed out that B-site was the key factor to catalyst for ABO_3 . For LaMO_3 , the oxidative (Cr, Mn) and reductive (Co, Ni, and Rh) nonstoichiometry perovskites were correlated with its catalytic performance. Using density functional theory, Liu et al. [13] suggested that the band gap is decreased by the doping Mg in $\text{LaFe}_{1-x}\text{Mg}_x\text{O}_3$, which increases the adsorption energy of O_2 . Compared with extensive studies for the effects of different metal cations in B-site of LaBO_3 on VOCs combustion, the effects of cations in A-site of LaBO_3 have seldom been reported. Recently, Pecchi et al. [14] reported that perovskite-type $\text{La}_{1-x}\text{Ca}_x\text{FeO}_3$ had the highest catalytic activity for methane combustion when x is equal to 0.4. They ascribed the high activity of $\text{La}_{1-x}\text{Ca}_x\text{FeO}_3$ to the formation of Fe^{4+} and oxygen vacancies. Does this conclusion still work for the cation with high valence in A-site, such as Ce^{4+} ion?

To explore this question, the $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ compounds, in which the A-site La atom was partially substituted by Ce atom were systematically studied in the present work. Using a sol-gel method, a series of perovskite-type $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ with different x ratio were synthesized. Then, the physical structures and chemical properties were characterized by X-ray diffraction spectrometer (XRD), BET surface area measurements, scanning electron microscopy (SEM) images, and temperature-programmed reduction (TPR). The

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corresponding performances of catalytic combustion of methane for $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ catalysts were evaluated.

2. Experimental

2.1. Catalyst preparation

The catalysts were prepared by a sol-gel method, which is chosen here due to its easiness to operate, relatively low calcination temperature combined with small and uniform sizes of as-synthesized products. $\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ and $\text{La}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ with different molar ratios of $x/(1-x)$ ($x = 0, 0.1, 0.2, 0.3, 0.4$, and 0.5) were dissolved in deionized water, and mixed with a $\text{Fe}(\text{NO}_3)_3$ solution at a constant composition level. To this solution, the citric acid with a acid/(Ce + La) molar ratio of 1.5/1 was added. After fully mixing, a NH_4OH solution was introduced and stirred at 80°C in a water bath for 30 min. Then the mixture was dried at 200°C for 1 h in an oven, followed by calcination at 750°C for 4 h in a muffle furnace. The final mixed oxides were composed of $(1-x)\text{La}/x\text{Ce}/\text{Fe}$ ($x = 0, 0.1, 0.2, 0.3, 0.4$, and 0.5) in molar ratio, respectively.

2.2. Catalyst characterization

X-ray diffraction (XRD) patterns were collected on a PANalytical X'Pert PRO MPD powder diffractometer using $\text{Cu K}\alpha$ radiation ($\lambda = 0.1542 \text{ nm}$, $40 \text{ kV}/40 \text{ mA}$). The patterns were collected in a 2θ range from 10° to 110° . The lattice parameter and the mean crystallite size were determined by the Rietveld method using JADE 6.5 software. Brunauer–Emmett–Teller (BET) surface areas were measured by N_2 adsorption on a Quantachrome Autosorb-1 (Quantachrome Instruments). SEM micrographs were recorded by the field emission scanning electron microscope (Hitachi S-4800).

Hydrogen temperature-programmed reduction (H_2 -TPR) measurements were performed in an atmospheric quartz tube flow reactor with a thermal conductivity detector (TCD). Under a flow of 30 ml min^{-1} 5 vol.% H_2/N_2 , the samples were heated to 900°C at a rate of $10^\circ\text{C}/\text{min}$.

2.3. Catalyst testing

The catalytic combustion of methane was carried out in a quartz tubular (i.d. = 6 mm) fixed-bed reactor under atmospheric pressure. 200 mg of the catalyst (150–180 μm) was loaded in the reactor, and the reaction temperature was detected by a thermocouple placed in the middle of the catalyst bed. The flow rate of gas reactant was 40 ml min^{-1} , corresponding to a space velocity of $12,000 \text{ ml g}_{\text{cat}}^{-1} \text{ h}^{-1}$. The reactant is a gas mixture of 0.5 vol.% CH_4 , 3 vol.% O_2 and 96.5 vol.% N_2 . The CH_4 concentrations before and after oxidation were analyzed using a Shimadzu GC-14 gas chromatograph equipped with a FID detector and a Supelcowax-10 ($30 \text{ m} \times 0.25 \text{ mm} \times 0.25 \mu\text{m}$) column. Conversion of CH_4 was defined as the following equation:

$$\text{CH}_4 \text{ conversion} = \frac{[\text{CH}_4]_{\text{in}} - [\text{CH}_4]_{\text{out}}}{[\text{CH}_4]_{\text{in}}} \times 100\%$$

where $[\text{CH}_4]_{\text{in}}$ and $[\text{CH}_4]_{\text{out}}$ are the methane concentration in the feed gas and product, respectively.

3. Result

3.1. Structural characterization

As shown in Fig. 1, the XRD diffractograms of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ perovskite with different Ce/La ratio calcined at 750°C , the characteristic diffraction peaks ($2\theta = 32.4^\circ, 46.6^\circ$, and 57.9°) have been

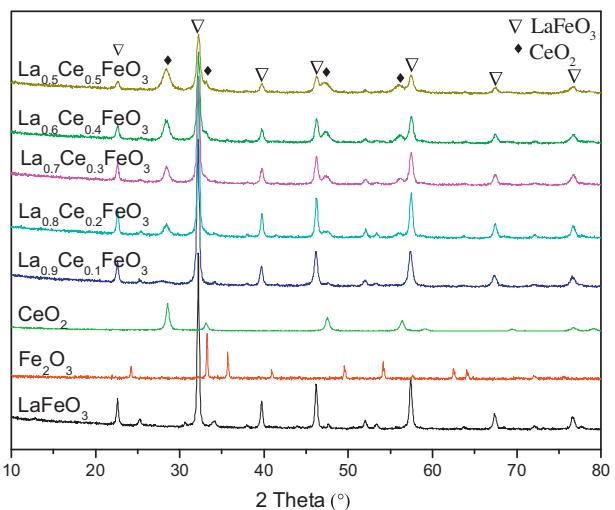


Fig. 1. XRD patterns of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$, CeO_2 and Fe_2O_3 samples.

observed corresponding to the orthorhombic structure of LaFeO_3 with a space group of $Pnma$, N.62. When x is 0.1, no XRD diffraction peaks of ceria are detected, indicating that the doped Ce atoms might have embodied into the LaFeO_3 lattice to form a solid solution. When x increases to 0.2, small diffraction peaks of cubic ceria appear. With the increase of x , the peak intensity of ceria increases, and the peak intensity of LaFeO_3 decreases, simultaneously. This implies the formation of ceria phase and its separation from perovskite oxide. Similar phase separation has been observed for the $\text{La}_{1-x}\text{Ca}_x\text{FeO}_3$ perovskite when x is higher than 0.4 [14]. Although no Fe_2O_3 peaks were detected in the XRD diffraction spectroscopy, Fe_2O_3 is proposed to be separated from perovskite oxides accompanying with the separation of CeO_2 phase, which will be proved by the following H_2 -TPR profiles.

Table 1 lists the BET surface areas and lattice parameters of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ with different x values. All the catalysts have been calcined at 750°C . It can be learned from Table 1 that the surface area of $\text{La}_{0.9}\text{Ce}_{0.1}\text{FeO}_3$ is almost the same as that of pure LaFeO_3 . This might be due to the fact that Ce atoms embody into the A-site without changing the pristine perovskite structure. This is consistent with the above XRD results, of which $\text{La}_{0.9}\text{Ce}_{0.1}\text{FeO}_3$ and LaFeO_3 show the same diffraction patterns. With the further increase of x , the surface area of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ firstly increases from $9.7 \text{ m}^2/\text{g}$ in LaFeO_3 to $13.3 \text{ m}^2/\text{g}$ when $x = 0.2$ and reaches $17.7 \text{ m}^2/\text{g}$ when $x = 0.3$, then slightly decreases to about $15 \text{ m}^2/\text{g}$ when x equals to 0.4 and 0.5. The increase in surface area of $\text{La}_{0.8}\text{Ce}_{0.2}\text{FeO}_3$ might be caused by the separation of CeO_2 and Fe_2O_3 nano-particles from perovskite oxide. When x is larger than 0.3, the separated oxide particles might begin to grow slightly and lead to the corresponding decrease in surface area. This finding is consistent with the XRD results i.e. the intensity of diffraction peak increases with increasing x in $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ and will be further verified by SEM measurements.

Table 1
BET surface areas and lattice parameters $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$.

Catalysts	$S_{\text{BET}}/\text{m}^2/\text{g}$	Lattice constants/ \AA			$V/\text{\AA}^3$	Crystallite size/nm
		a	b	c		
LaFeO_3	9.7	5.553	7.867	5.563	243.02	37.0
$\text{La}_{0.9}\text{Ce}_{0.1}\text{FeO}_3$	9.1	5.567	7.855	5.533	241.95	32.6
$\text{La}_{0.8}\text{Ce}_{0.2}\text{FeO}_3$	13.3	5.565	7.855	5.556	242.87	33.6
$\text{La}_{0.7}\text{Ce}_{0.3}\text{FeO}_3$	17.7	5.567	7.855	5.533	241.95	30.0
$\text{La}_{0.6}\text{Ce}_{0.4}\text{FeO}_3$	14.5	5.567	7.855	5.533	241.95	28.4
$\text{La}_{0.5}\text{Ce}_{0.5}\text{FeO}_3$	15.0	5.567	7.855	5.533	241.95	25.2

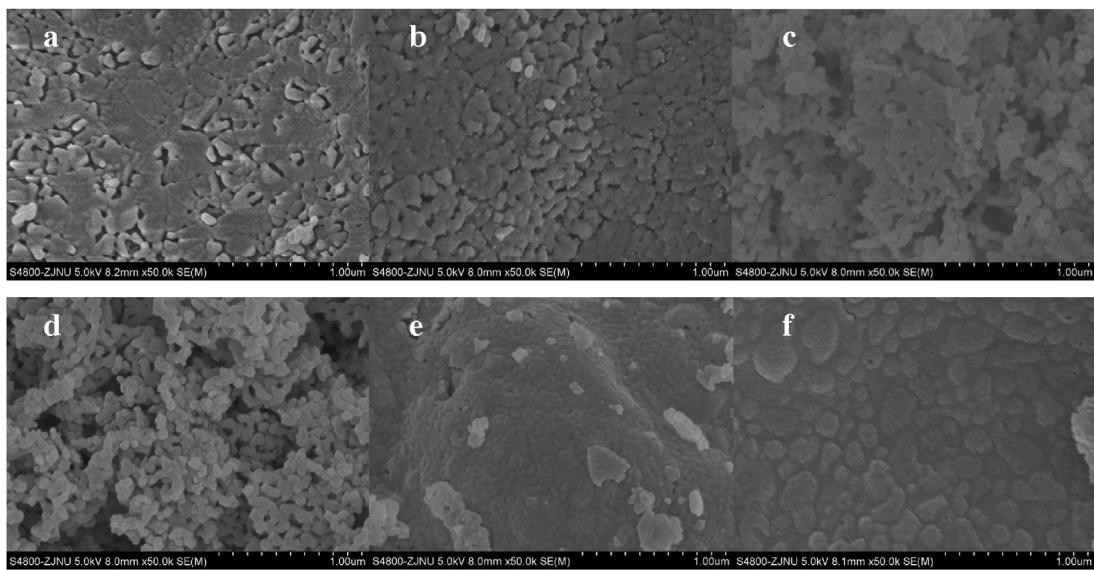


Fig. 2. SEM images of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ samples. (a)–(f) corresponds to the x value of 0, 0.1, 0.2, 0.3, 0.4, and 0.5, respectively.

According to the SEM images of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ (Fig. 2), the surface morphology of LaFeO_3 and $\text{La}_{0.9}\text{Ce}_{0.1}\text{O}_3$ is similar. However, the small branch-shape structures were observed when x increases to 0.2 and 0.3, which might be attributed to the phase separation of ceria and Fe_2O_3 from perovskite oxide. When x is up to 0.4 and 0.5, relatively big nano-particles were obtained due to the increase of separated ceria and Fe_2O_3 . The SEM results support the above analysis results of XRD and BET surface area.

The redox properties of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ with different molar ratios of La/Ce were studied by H_2 -TPR, as shown in Fig. 3. It can be seen from Fig. 3 that a broad reduction peak is observed for pure LaFeO_3 at c.a. 500 °C; while a shoulder peak appears at 430 °C for $\text{La}_{0.9}\text{Ce}_{0.1}\text{O}_3$, which might be ascribed to the effects of doped Ce atoms. When x increases to 0.2, a new peak is observed at 390 °C for $\text{La}_{0.8}\text{Ce}_{0.2}\text{O}_3$, which is attributed to the reduction of Fe_2O_3 . For $\text{La}_{0.7}\text{Ce}_{0.3}\text{O}_3$, the peak area at 390 °C increases greatly compared with $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$ ($x=0, 0.1$ and 0.2). Simultaneously, a new β peak at 500 °C appears, which is attributed to the reduction of small ceria nano-particles separated from perovskite oxide based on the above XRD and nitrogen adsorption results. With the further increase of x , the two reduction peak areas increase and peak position slightly

shifts toward high temperature due to the enhanced size of oxide particles.

3.2. Catalytic activity

Fig. 4 shows the catalytic activities of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$, CeO_2 and Fe_2O_3 samples for methane combustion. It can be seen from Fig. 4 that pure ceria has a low CH_4 combustion activity, of which the temperatures of CH_4 conversion at 10% and 50% (T_{10} and T_{50}) are 450 and 560 °C, respectively; the oxidative activity of LaFeO_3 is higher than that of ceria at low temperature with T_{10} of 390 °C, while the T_{10} and T_{50} of Fe_2O_3 are 410 and 520 °C, respectively. When a small quantity of Ce atom was doped into LaFeO_3 to form the solid solution, $\text{La}_{0.9}\text{Ce}_{0.1}\text{O}_3$, the perovskite sample's reactivity improves greatly with the reduction of T_{10} and T_{50} to 360 and 460 °C, respectively. With the increase of x , the oxidative activity of methane increases slightly first and reaches the maximum when $x=0.3$. The corresponding T_{90} is as low as 510 °C. However, when x is further raised to 0.4 and 0.5, its activity of CH_4 combustion decreases, correspondingly.

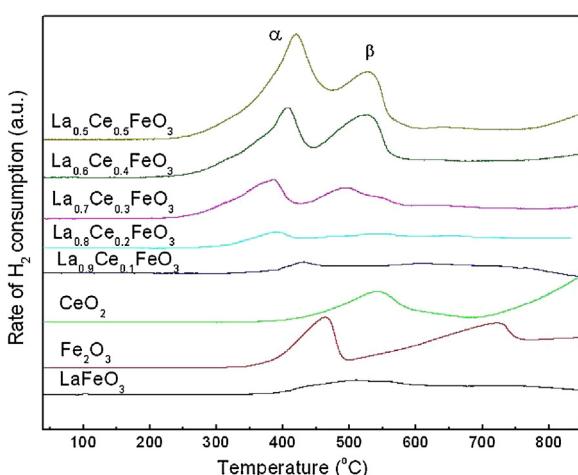


Fig. 3. H_2 -TPR profiles of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$, CeO_2 and Fe_2O_3 samples powder samples.

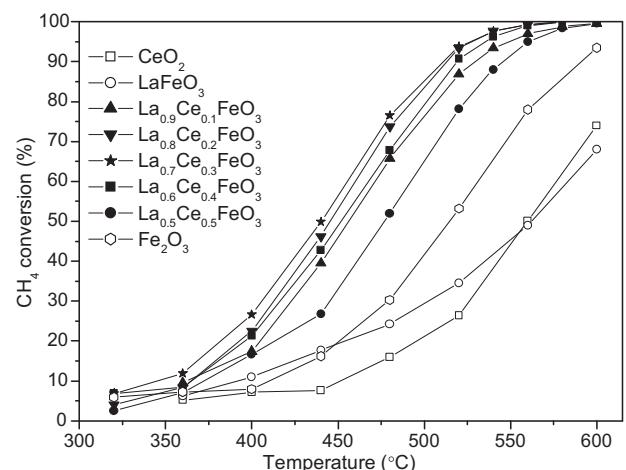


Fig. 4. Catalytic activities of $\text{La}_{1-x}\text{Ce}_x\text{FeO}_3$, CeO_2 and Fe_2O_3 samples for methane combustion.

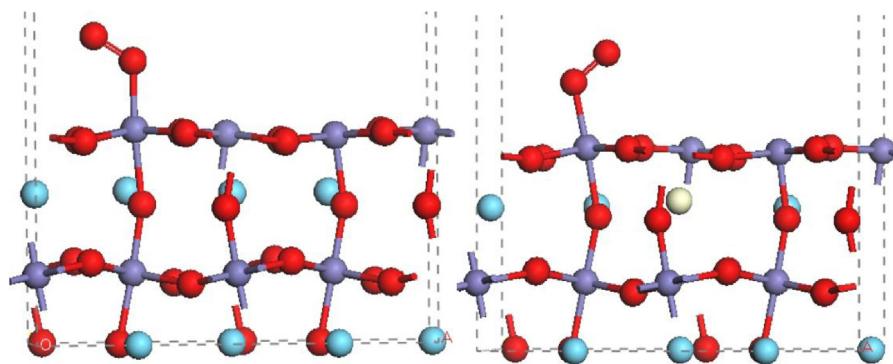


Fig. 5. O₂ adsorbed on LaFeO₃(010) and La_{0.875}Ce_{0.125}FeO₃(010) surfaces. ● : La, ○ : Ce, ● : Fe, ● : O.

Table 2

Bader charges of Fe, Ce and O₂ on LaFeO₃(010) and La_{0.875}Ce_{0.125}FeO₃(010).

Atomic type	LaFeO ₃	La _{0.875} Ce _{0.125} FeO ₃	O ₂ /LaFeO ₃	O ₂ /La _{0.875} Ce _{0.125} FeO ₃
Fe _{total}	13.34	13.14	13.47	13.41
Ce	–	2.20	–	2.18
O _{2,ads}	–	–	-0.43	-0.39

Note: Bader charge of Ce⁴⁺(CeO₂) and Ce³⁺(Ce₂O₃) ions are 2.29 and 1.86 calculated herein, respectively.

4. Discussion

The great improvement of CH₄ combustion for La_{0.9}Ce_{0.1}FeO₃ might be due to the doping effects of Ce. When Ce⁴⁺ ion substitutes the position of La³⁺ in A-site, it might cause the charge imbalance of perovskite, which was also observed in La_{1-x}Ca_xFeO₃ and La_{1-x}Sr_xFeO₃ [10,14]. Until now, it was widely accepted that the doped metals with a lower valence (Ca²⁺ or Sr²⁺) compared with La³⁺ might lead to more oxygen vacancies in the perovskite oxides [1,8]. Then, O₂ might fill into the surface vacancies and be activated to form superoxide or peroxide species [15]. Correspondingly, the activated oxygen species take part into the successive oxidative reactions. However, the doped Ce⁴⁺ ion is quite different from the metal ions with low valence then how does it affect the oxidative performances of La_{1-x}Ce_xFeO₃?

To explore the possible effects of doped Ce ion in La_{1-x}Ce_xFeO₃, we calculate O₂ adsorption behaviors on the LaFeO₃(010) and La_{0.875}Ce_{0.125}FeO₃(010) surfaces using density functional theory method. The corresponding optimized structures of O₂ were shown in Fig. 5. Detailed descriptions of theoretical model and method can be seen in the supplementary materials.

As shown in Fig. 5(a), O₂ stands on the top site of surface Fe for LaFeO₃(010) to form a Fe—O bond with the length of 1.917 Å. The O—O bond of adsorbed O₂ is elongated to 1.282 Å, which is longer than that of free O₂ (1.21 Å). The corresponding adsorption energy is -0.42 eV, which is consistent with the value reported in the literature [13,16]. Similar structure was also obtained for O₂ on La_{0.875}Ce_{0.125}FeO₃(010) surface. Its Fe—O bond length is 1.889 Å, which is slightly shorter than on LaFeO₃. While the O—O bond (1.289 Å) is slightly longer than on LaFeO₃, indicating relatively strong interactions between O₂ and the doped perovskite oxides. This conclusion is further proved by the adsorption energy of O₂ on La_{0.875}Ce_{0.125}FeO₃(010) (-0.69 eV), which is higher than that on LaFeO₃.

To further understand the effects of doped Ce ion, the bader charges of Ce, O₂ and the sum of Fe ions were listed in Table 2. The total charges of La and O ions almost did not change no matter with and without Ce doped and were not listed herein. As seen in Table 2, the charge of Ce ion in La_{0.875}Ce_{0.125}FeO₃ is 2.20, which is much higher than that of Ce³⁺(1.86) and approaches to Ce⁴⁺(2.29)

ion, indicating that the doped Ce ion approximates Ce⁴⁺. Due to the introduction of Ce⁴⁺ ion, the total charge of Fe ions decreases by 0.2 to keep La_{0.875}Ce_{0.125}FeO₃ electronic neutral. Therefore, the electrons of Fe ions increase and stronger interactions occur between Fe and the adsorbed O₂. Correspondingly, O₂ is activated and might lead to higher oxidative reactivity on La_{1-x}Ce_xFeO₃ than on pure LaFeO₃, which is observed by the catalytic test for methane combustion in the present work.

5. Conclusion

A series of La_{1-x}Ce_xFeO₃ ($x=0\text{--}0.5$) perovskite oxides were prepared, characterized and tested by methane catalytic combustion reaction. Our results show that the activity of LaFeO₃ was highly improved due to the introduction of Ce in the A-site of the perovskite catalysts, and reaches the maximum for La_{0.7}Ce_{0.3}FeO₃. The corresponding T_{90} is as low as 510 °C. Density functional theory calculation suggests that the electrons of Fe ions increase in La_{0.875}Ce_{0.125}FeO₃ due to the introduction of Ce⁴⁺ ion, and leads to stronger interactions with O_{2,ads}. Therefore, the adsorption energy of O₂ on La_{0.875}Ce_{0.125}FeO₃ increases and the O—O bond is activated. Correspondingly, the Ce doped perovskites have higher oxidative activity than pure LaFeO₃.

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Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.apusc.2013.03.091>.

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